

Biopolymer-assisted construction of porous SnO₂ microspheres with enhanced sensing properties

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ABSTRACT

Nanoparticle-assembled SnO₂ microspheres with a porous structure are prepared by a biopolymer-assisted synthetic method. The biopolymer used herein is sodium alginate. Besides sodium alginate, an ethylene glycol–water mixed solvent system is also proven to be necessary for the formation of the porous SnO₂ microspheres. The gas-sensing properties of this material are evaluated by a range of gases, and the results show that the gas sensor based on the porous SnO₂ material exhibits the highest sensing response towards ethanol relative to other testing gases. The sensing response of the porous SnO₂ is ~3 and 4 times as high as those of the solid SnO₂ microspheres (which are prepared in the absence of sodium alginate) and the commercially available SnO₂ nanoparticles, respectively. In addition, the porous SnO₂ sensor exhibits a wide response range from 0.5 to 200 ppm for ethanol detection. In the testing range from 0.5 to 200 ppm, the logarithm of the response value shows a good linear dependency on the logarithm of ethanol concentration, indicating that the porous SnO₂ sensor may be used for quantitative detection of ethanol vapor.

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1. Introduction

Gas sensors based on semiconducting metal oxides have been extensively used to detect a wide range of gases and organic vapors. The fundamental mechanism of semiconductor sensors primarily depends on the interaction between the chemisorbed oxygen species on the sensor surface and the gas molecules to be detected, which results into observable variation in the electrical resistance of the sensing material. Undoubtedly, the rapidly expanding fields of nanoscience and nanotechnology have accelerated the development of semiconductor sensors [1]. Many materials at nanoscale often show superior sensing properties (e.g., sensitivity, selectivity and/or reliability) to their bulk counterparts, because of their structural advantages such as high surface area and large density of surface reactive sites. Despite many achievements made in the development of various sensing nanomaterials, there are still many challenges ahead. Some of them include (i) large-scale and “green” preparation of nanostructures with a special

morphology/composition; (ii) deep understanding of structure–property correlations for a particular sensing nanomaterial; (iii) goal-directed design/engineering of effective sensing nanostructures. This, hence, calls for more research efforts in the nanosensor technology—a burgeoning and fascinating area.

Tin dioxide (SnO₂) – an important n-type wide-band gap semiconductor – is one of promising oxide materials that can be employed in wide gas-sensing applications [2]. In order to improve SnO₂’s sensing properties, bulk doping with heteroatoms [3–5] and surface modification with noble metal particles [6–8] or another semiconductors [9,10] were attempted in the past several decades. By comparison, nanostructuring SnO₂ material is a simpler and more cost-effective strategy, and it might also avoid the high cost of noble metals and/or the complexity of multicomponent systems. A considerable amount of research effort, therefore, has been devoted to prepare a variety of SnO₂ nanostructures for sensing applications [11–22]. In particular, hierarchical SnO₂ nanostructures with a porous structure have attracted extensive attention as an important family of sensing materials in recent years [23–30], because of their large surface area, low density and unique morphological feature. For example, with the assistance of cetyltrimethylammonium bromide as the structure-directing agent, Kuang et al. synthesized mesoporous SnO₂ nanospheres and used them as good sensing

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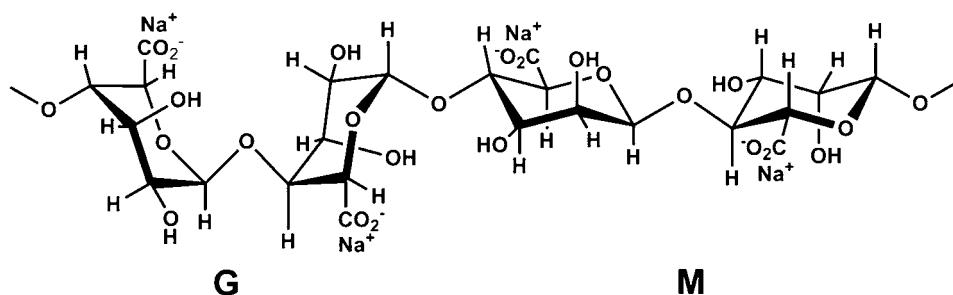


Fig. 1. Structure of sodium alginate containing guluronate (G) and manuronate (M) units.

materials [23]. Using glucose as the additive, Manjula et al. reported the successful synthesis of porous SnO_2 nanospheres with excellent gas sensing performances for hydrogen detection [24]. In addition, Xie et al. synthesized a porous SnO_2 sensing material in the presence of sodium dodecyl benzenesulfonate (SDBS) and terephthalic acid [26]. Weimar et al. synthesized a SnO_2 porous nanomaterial in a water-in-oil microemulsion system [27], which was established using cetyltrimethylammoniumbromide (CTAB) as a surfactant, hexanol as a cosurfactant and *n*-dodecane as the nonpolar oil phase.

In this paper, we report a facile synthetic route to porous SnO_2 microspheres (p- SnO_2) using biopolymer sodium alginate (NaAlg) as the structure-directing agent for the first time. The as-obtained p- SnO_2 nanomaterial can serve as a highly sensitive material for ethanol detection, thanks to its novel hierarchical structure and porous texture. Notably also, this p- SnO_2 nanomaterial is among the most sensitive SnO_2 -based ethanol sensing materials (Table S1, ESI).

Sodium alginate is a naturally available carbohydrate polymer that is composed of manuronate acid (M-block) and guluronic acid (G-block) units (Fig. 1); and it is also considered as a “green” chemical agent because it is biodegradable and non-toxic. To the best of our knowledge, although sodium alginate has already been used as stabilizer or structure-directing agent for the synthesis of metal and metal oxide nanoparticles [31–38] as well as several metal (hydro)oxide hollow structures [39–43], there were no reports on the biopolymer (sodium alginate)-assisted synthesis of porous SnO_2 nanostructures previously.

2. Experimental

2.1. Reagents and materials

Sodium alginate was purchased from Tianjin Guangfu Fine Chemical Research Institute. Sodium stannate trihydrate ($\text{Na}_2\text{SnO}_3 \cdot 3\text{H}_2\text{O}$) was purchased from Tianjin Fuchen Chemical Reagent Factory. Ethanol and ethylene glycol were purchased from Beijing Chemical Works. The reference SnO_2 sample was purchased from Sinopharm Chemical Reagent Co., Ltd. All the above chemicals were used without further purification. Deionized water was used through the experiments.

2.2. Biopolymer-assisted synthesis of porous SnO_2 microspheres

In a typical synthesis of porous SnO_2 material, $\text{Na}_2\text{SnO}_3 \cdot 3\text{H}_2\text{O}$ (1.34 g, 5 mmol) was added into ethylene glycol (25 mL) to form solution 1; and sodium alginate (0.25 g) was added into H_2O (25 mL) to form solution 2. Solutions 1 and 2 were mixed under stirring and the resulting mixture was put into a 100 mL Teflon-sealed autoclave, which was then heated at 180 °C for 24 h. After cooled to room temperature, the resulting white precipitate was centrifuged and thoroughly washed with deionized water and ethanol several times, and dried in air at 80 °C. The finally-obtained solid product is

the porous SnO_2 material, and this material designated as p- SnO_2 . For comparison, the other solid SnO_2 microspheres sample was prepared in the absence of sodium alginate, under otherwise similar synthetic procedures as above. The solid SnO_2 microspheres sample was designated as s- SnO_2 .

2.3. General characterization

The powder X-ray diffraction (XRD) patterns were performed with a Rigaku D/Max 2550 X-ray diffractometer using $\text{Cu K}\alpha$ radiation ($\lambda = 1.5418 \text{ \AA}$) operated at 200 mA and 50 kV. The scanning electron microscopic (SEM) images were carried out on a JEOL JSM 6700F electron microscope. The transmission electron microscopy (TEM) and high-resolution TEM (HRTEM) images were obtained on a Philips-FEI Tecnai G2S-Twin. BET surface area and BJH pore diameter were measured by using a Micromeritics ASAP 2020M system.

2.4. Sensor fabrication and testing

The gas sensor was fabricated by pasting viscous slurry of the obtained sample onto a ceramic tube with a diameter of 1 mm and a length of 4 mm, which was positioned with a pair of Au electrodes and four Pt wires on both ends of the tube. A Ni-Cr alloy coil passing through the tube was employed as a heater to control the operating temperature. The schematic structure of the sensor is shown in Fig. 2. For comparison, all of the sensors were fabricated using the same method, and only the difference may be the SnO_2 sensing material. Gas sensing tests were performed on a commercial CGS-8 Gas Sensing Measurement System (Beijing Elite Tech Company Limited).

Gas sensing properties were measured using a static test system which included a test chamber (~1 L in volume). Environmental air with a relative humidity of ~30% was used as both a reference gas and a diluting gas to obtain the desired concentrations of target gases. A typical testing procedure was as follows. After the target gas was injected into the test chamber for about 5 min by a microsyringe, the sensor was put into the test chamber. When the

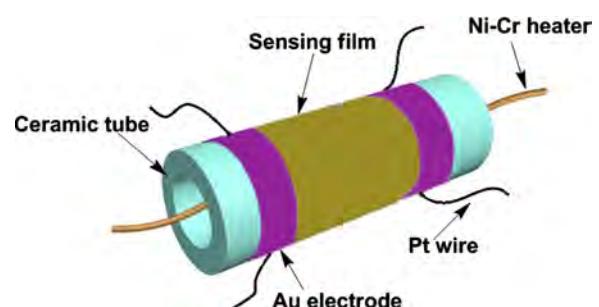


Fig. 2. Schematic diagram showing the structure of a typical gas sensor.

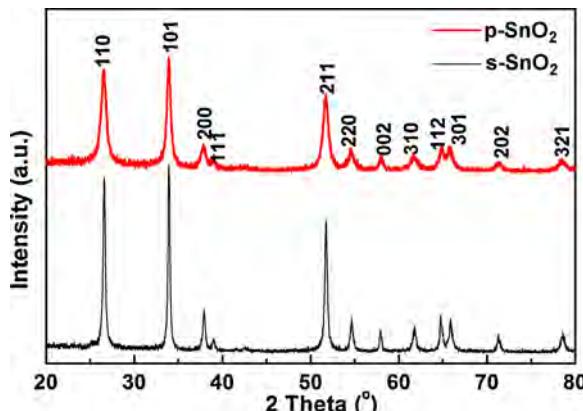


Fig. 3. XRD patterns of p-SnO₂ and s-SnO₂.

response reached a constant value, the sensor was taken out to recover in fresh air. The sensor response is defined as the ratio of R_a to R_g , where R_a and R_g are the electrical resistance of the sensor in atmospheric air and in the testing gas, respectively.

3. Results and discussion

The porous microspheres (p-SnO₂) were hydrothermally synthesized with sodium alginate as the structure-directing agent in an ethylene glycol-water mixed solvent system ($v/v = 1:1$). Firstly, when sodium alginate was not added in the above mentioned reaction, only nonporous SnO₂ microspheres (i.e., s-SnO₂) were obtained (see below), demonstrating the importance of sodium alginate for the formation of porous structure. Secondly, suitable amount of water in the reaction system was required because sodium alginate can be dissolved in water, but not in ethylene glycol. Thirdly, the presence of ethylene glycol was also very important because when ethylene glycol was not added in the above reaction system, only aggregated SnO₂ nanoparticles (Fig. S1 in ESI) were obtained, indicating that ethylene glycol might be helpful for the

self-assembly of SnO₂ nanoparticles. Similar phenomena were also observed in some previous reports on the preparation of hierarchical oxide/hydroxide nanostructures [44,45]. Although the synthesis of the porous SnO₂ microspheres originates from such a multi-component system, this synthetic approach is facile, controllable and reproducible.

For comparative purpose, s-SnO₂ (which was prepared in the absence of sodium alginate) was also fully characterized, along with p-SnO₂. The XRD patterns of p-SnO₂ and s-SnO₂ are shown in Fig. 3. Both XRD patterns can be well indexed to the tetragonal phase of SnO₂ (JCPDS card No. 41-1445). No XRD peaks of impurities were detected. Moreover, the XRD peaks of p-SnO₂ appear broader than those of s-SnO₂; this indicates that the primary particles in p-SnO₂ are smaller in size than those in s-SnO₂. The crystal sizes of p-SnO₂ and s-SnO₂ were estimated by using the Scherrer formula to be around 9 nm and 21 nm, respectively.

The morphologies of p-SnO₂ and s-SnO₂ were observed by scanning electron microscopy (SEM). The low-magnification SEM image (Fig. S2, ESI) shows that p-SnO₂ consists of spherical particles with a diameter of 200–400 nm. The more detailed structural feature of p-SnO₂ was examined from high-magnification SEM image (Fig. 4A). It is clearly seen that the surface of p-SnO₂ particles is composed of densely packed nanoparticles with several nanometer in size. In addition, some microspheres with a small hollow structure were also seen through some broken particles (see SEM images in Fig. 4B and Fig. S3, ESI). Furthermore, s-SnO₂ also possesses nearly spherical microparticles, but the size of them is much larger than that of p-SnO₂ (Fig. 4C and D). From the above results, it is presumable that the presence of sodium alginate could alter the assembly pattern of SnO₂ nanoparticles.

The transmission electron microscopy (TEM) was used to study the structure of p-SnO₂ and to reveal the structural differences between p-SnO₂ and s-SnO₂. The TEM images of p-SnO₂ (Fig. 5A and Fig. S4, ESI) show a slight lighter region at the center and a continuous darker region around the edges of the microspheres, indicating that the lower particle density at the center of the p-SnO₂ microspheres. In contrast, s-SnO₂ shows a solid structure, demonstrating the importance of sodium alginate in controlling the self-assembly

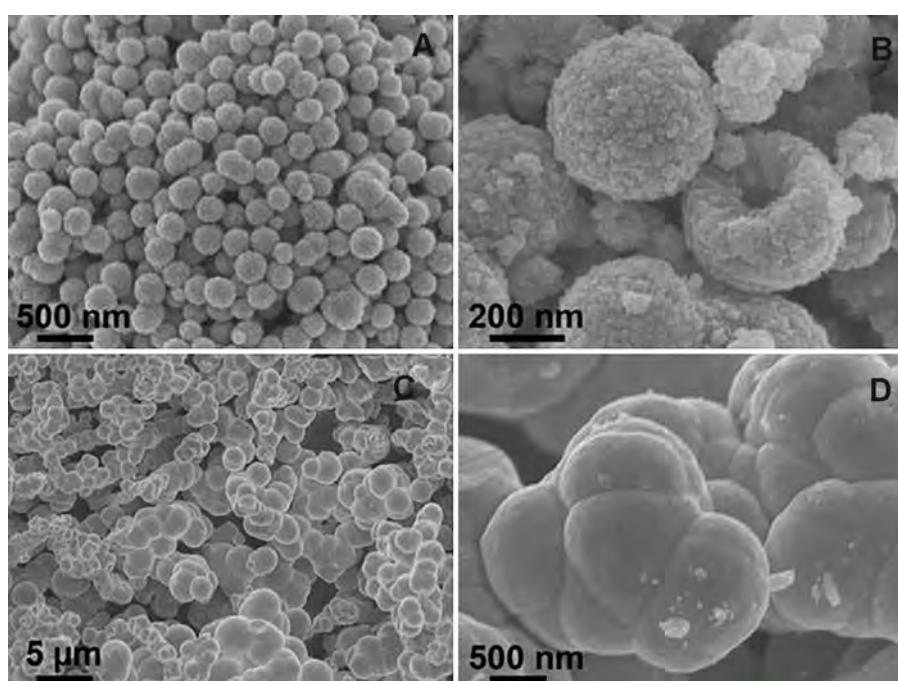


Fig. 4. SEM images of (A and B) p-SnO₂ and (C and D) s-SnO₂.

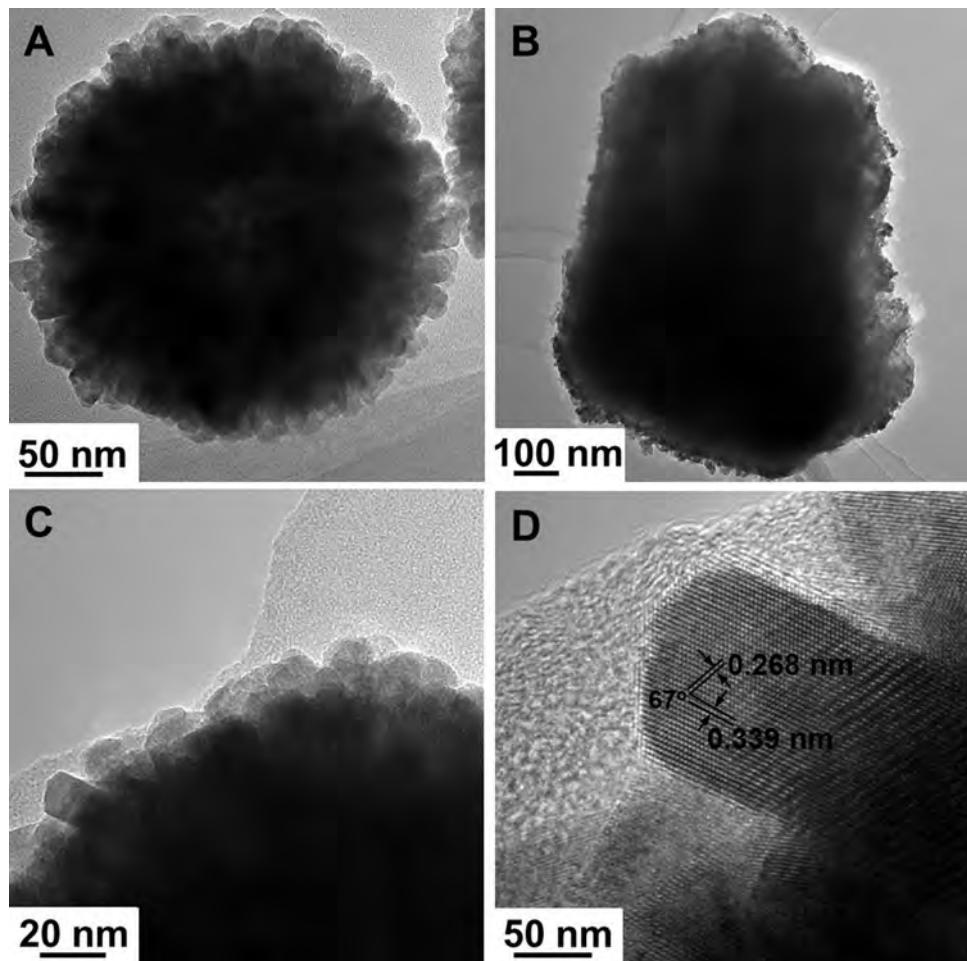


Fig. 5. TEM images of (A) p-SnO₂ and (B) s-SnO₂; and (C and D) TEM and HRTEM images of p-SnO₂.

behavior of SnO₂ nanoparticles. The high-resolution TEM (HRTEM) image of the edge of p-SnO₂ (Fig. 5C) shows the shell to be composed of tightly interconnected SnO₂ nanocrystals of ~10 nm in size. The lattice spacing of the SnO₂ nanocrystals (Fig. 5D) was found to be ~0.268 and 0.339 nm, which is consistent with (101) and (110) planes of tetragonal SnO₂.

The N₂ adsorption-desorption isotherms of p-SnO₂ (Fig. 6) exhibit a characteristic type-IV isotherm with an H3 hysteresis loop, indicating the presence of mesoporous/macroporous structure in the material. The corresponding BJH pore-size distribution derived from desorption branch of the isotherm shows a wide pore-size distribution centered at 15 nm, further confirming the existence of mesopores in the material. The corresponding BET surface area of this material is about 29 m² g⁻¹, which is large than that (9 m² g⁻¹) of s-SnO₂.

Because of its advantageous structural features such as porous structure, it is expected that p-SnO₂ would serve as a good gas sensing material (see the sensor fabrication and testing in experimental section). For comparison, two more gas sensors were fabricated from s-SnO₂ and the commercial SnO₂ nanoparticles with a BET surface area of 6 m² g⁻¹ (Com-SnO₂, see the SEM image and the XRD pattern in Fig. S5, ESI). The sensing performance of the SnO₂ materials was evaluated by using ethanol as the testing gas, due to the importance of ethanol sensors in food control applications. The response and recovery times of the sensor are defined as the times taken by the sensor to achieve 90% of the total resistance change. The optimal operating temperature of the SnO₂ sensor was determined by testing 200 ppm ethanol, and the result shows that

the optimal operation temperature of the sensor is around 170 °C (Fig. 7). This value is within the scope of temperatures that are usually required to make the semiconductor-based sensors work well, and this operating temperature was applied in all the sensing tests. Note that the target gas was maintained at room temperature (~25 °C) during the whole testing process.

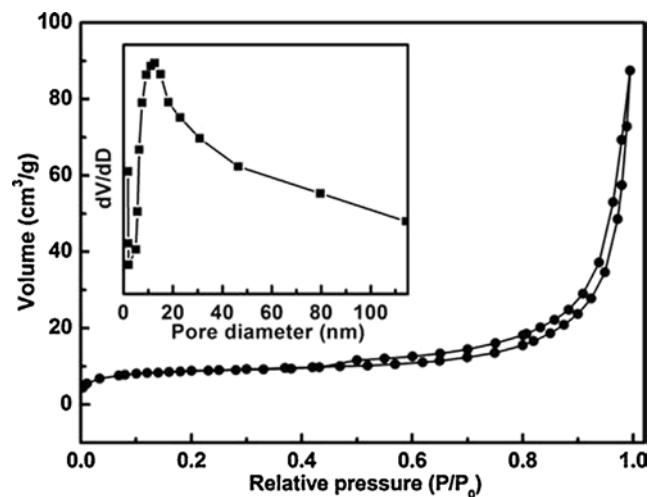


Fig. 6. N₂ adsorption/desorption isotherms for the p-SnO₂ sample with the corresponding pore size distribution shown in the inset.

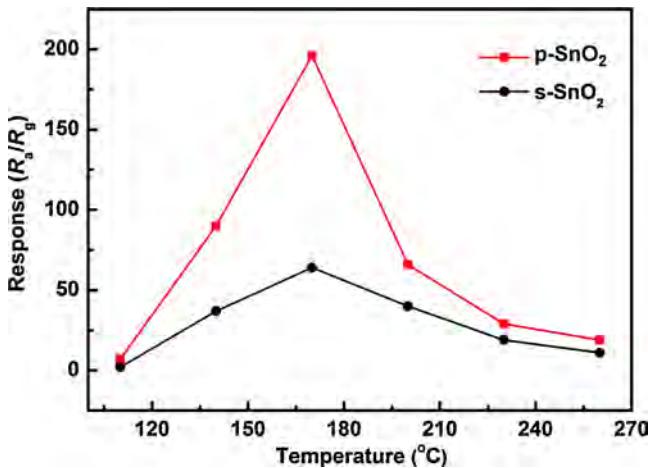


Fig. 7. Response of the sensors based on p-SnO₂ and s-SnO₂ as a function of the operating temperature for the detection of ethanol with a concentration of 200 ppm.

Fig. 8A shows the typical dynamic response-recovery curve of the p-SnO₂ sensor with increasing ethanol concentrations. It is seen that the sensor has a wide response range of 0.5–200 ppm for ethanol detection. The response increases significantly with the increase of ethanol concentration. Moreover, the response is about 1.5 at an ethanol concentration of 0.5 ppm, and thus the detection limit of the p-SnO₂ sensor can be said to be 0.5 ppm. For the concentrations of 0.5, 1, 2, 5, 10, 20, 50, 100, 200 ppm, the responses of p-SnO₂ are about 1.5, 2.5, 4.0, 10, 19, 33, 64, 119 and 197, respectively. In the testing range from 0.5 to 200 ppm, the logarithm

of the sensor response shows a good linear dependency on the logarithm of ethanol concentration (**Fig. 8B**), indicating that the p-SnO₂ sensor may be used as quantitative detection of ethanol vapor. The response times of p-SnO₂ to ethanol are about 25 s, 20 s, 16 s, 7 s, 3 s, 5 s, 2 s, 3 s, and 4 s, for 0.5, 1, 2, 5, 10, 20, 50, 100, 200 ppm, respectively. By comparison, the response times of s-SnO₂ are about 14 s, 6 s, 5 s, 1 s, 2 s, 1 s, 1 s and 1 s for 1, 2, 5, 10, 20, 50, 100, 200 ppm ethanol, respectively. Obviously, the p-SnO₂ exhibits a longer response time for ethanol detection. The stability of the p-SnO₂ sensor was measured by testing 10 ppm ethanol at 170 °C (Fig. S7). Ten almost identical signals were observed in rapid succession in the presence of 10 ppm ethanol, demonstrating the excellent stability of p-SnO₂ sensor. **Fig. 8C** shows the gas concentration-dependent responses of the sensors based on the p-SnO₂, s-SnO₂ and Com-SnO₂ materials. It is obvious that the p-SnO₂ sensor exhibits response distinctly higher than those of the s-SnO₂ and Com-SnO₂ sensors, demonstrating the importance of a hierarchical structure and porous texture for SnO₂ sensing materials. For example, the sensing response of p-SnO₂ is ~3 and 4 times as high as those of s-SnO₂ and Com-SnO₂, respectively, in the presence of 200 ppm ethanol (**Fig. 8C** and S6, ESI). Also noteworthy, our p-SnO₂ nanomaterial is among the most sensitive SnO₂-based ethanol sensing materials (see Table S1 in ESI); and only several SnO₂ materials exhibit higher response towards ethanol than p-SnO₂ [12,15,16,46]. The sensing performance of the p-SnO₂ sensor was further investigated by testing a wide range of gases and organic vapors (**Fig. 8D**). The experimental results show that the p-SnO₂ sensor is more sensitive to ethanol than other gases/vapors including methanol, acetone, formaldehyde, hydrogen, carbon monoxide, benzene, methane and methylamine. From the above sensing results, it is concluded that the p-SnO₂

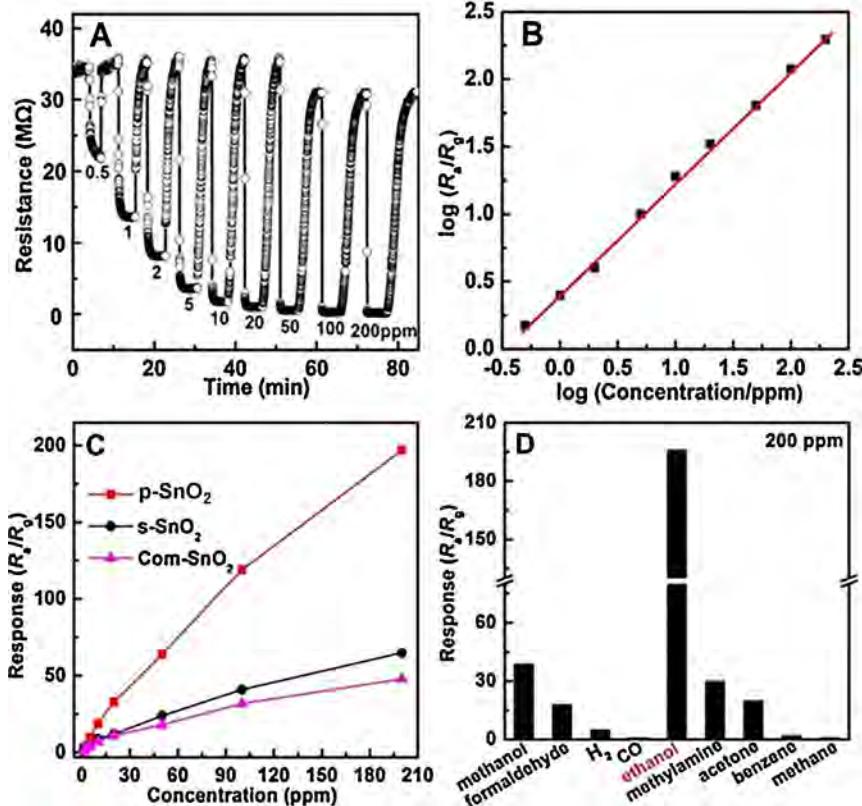


Fig. 8. (A) Dynamic response-recovery curves of the sensor based on p-SnO₂ for ethanol detection; (B) variation of the response (R) of the p-SnO₂ sensor with ethanol concentration (ppm); (C) comparison of the ethanol concentration-dependent responses of the p-SnO₂ sensor, the s-SnO₂ sensor and the Com-SnO₂ sensor; and (D) response comparison of the p-SnO₂ sensor in the presence of different test gases (200 ppm).

nanomaterial is very promising for the fabrication of an ethanol sensor because of its low operating temperature, high response, good selectivity and excellent stability.

4. Conclusions

In summary, we reported the biopolymer-assisted synthesis of porous SnO₂ nanomaterial, which was used as an excellent sensing material for ethanol detection. Both biopolymer sodium alginate and ethylene glycol-water mixed solvent system are necessary for the formation of the porous SnO₂ nanostructure. The significantly improved sensing performance is attributed to its novel hierarchical structure and porous texture. We believe that this study would further promote the exploitation of semiconductor nanomaterial for gas sensors.

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Appendix A. Supplementary data

Supplementary data associated with this article can be found, in the online version, at <http://dx.doi.org/10.1016/j.snb.2014.07.085>.

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